



K-1688

IN THE UNITED STATES PATENT AND TRADEMARK OFFICE

In the application of: Yeckley)
•) GROUP ART UNIT 1755
Serial No. 09/724,188)
Filed: November 28, 2000) Examiner: Group, Karl
For: SiAlON Containing Ytterbium)
And Method of Making	j ·

DECLARATION OF RUSSELL L. YECKLEY

- I, Russell L. Yeckley, hereby declare as follows that:
- 1. I am over the age of twenty-one (21) years and am a citizen of the United States of America and a resident of the Commonwealth of Pennsylvania.
- 2. I am the inventor of the above-captioned pending U.S. Patent Application, that I am an employee of Kennametal Inc., the assignee of the above-captioned patent application, and that I am making this Declaration on behalf of Kennametal Inc.
- 3. My post-high school educational background is as follows: in 1976 I received a B.S. in Ceramic Engineering from Penn State University and in 1984 I received a M.S. in Materials Engineering from the University of Pittsburgh.
- 4. Since 1979 I have been employed in various positions pertaining to the development and/or production of ceramic materials such as SiAlON materials and silicon nitride materials.
- 5. Under my direction I had performed some experiments wherein the compositions of the starting powder mixtures and the properties of the resultant SiAlON ceramics are set forth in Table 1 below.

Table 1. Composition and Properties of Sialon ab83612

	Composition	in Weight Pe	rcent						T		T	
	Si3N4	Si3N4	AIN Starck	Al2O3	Yb2O3	β Si3N4 Weight	Hvn		Kic		α sialon	Yb4SiAlO8N
	İ			Ceralox		Percent					Weight	
Batch	Starck M11	Ube SNE-03	Α	HPA-0.5	Molycorp		GPa	st dev	MPa m1/2			R.I.
4	21.70%	65.10%	4.5	2.5	6.2		18.24		7.05		31.5	3.3
1924D		86.80%	4.5	2.5	6.2	0	18.06	0.47	7.24	0.10	41.7	2.1
1924E	43.40%	43.40%	4.5	2.5	6.2		18.30		6.65		23.3	3.6

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Table 1 reports:

- (A) the starting compositions, selected physical properties, and selected phases present for Batch Nos. 1924C, 1924D and 1924E;
- (B) for each of the batches the starting powder components in weight percent wherein the components comprise Stark M11 silicon nitride powder (which has 8 weight percent beta silicon nitride), Ube SNE-03 silicon nitride powder (which has 0 weight percent beta silicon nitride), Stark Grade A aluminum nitride powder, Ceralox HPA-05 aluminum oxide powder, and Molycorp ytterbium oxide powder;
- (C) for each of the batches the beta silicon nitride content (weight percent) in the starting silicon nitride powder;
- (D) for each of the batches the Vickers hardness (18.5 kg load) reported in GPa (along with the standard deviation);
- (E) for each of the batches the fracture toughness (K _{IC}) determined by the Evans and Charles technique and reported in MPa•m ½ (along with the standard deviation);
- (F) for each of the batches the content of the alpha prime SiAlON phase in weight percent of the total of the alpha prime SiAlON phase and the beta prime SiAlON phase; and
- (G) for each of the batches the relative intensity of the crystalline intergranular phase Yb_4SiAlO_8N .
- 6. The powder mixtures as set forth in Table 1 were processed as follows: the compacts were belt-sintered in one atmosphere of flowing nitrogen for 30 minutes in each one of the following temperature zones: 1550 °C, 1700 °C, 1765 °C, and 1765 °C, and then hot isostatically pressed at a maximum temperature of 1830 °C for 30 minutes under a pressure of 20,000 pounds per square inch.

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- 7. Figures 1 and 2 attached hereto (Exhibit A) comprise backscatter scanning electron microscope (SEM) photomicrographs (with a 40 micrometer legend) of the polished surface wherein these two photomicrographs compare the resulting microstructure of Batch Nos. 1924E and 1924D respectively. In Figures 1 and 2 the brightest phase is the ytterbia aluminum silicon oxy-nitride grain boundary phase, the medium gray grains are the alpha prime SiAlON phase, and the darkest grains are the beta prime SiAlON phase.
- 8. A comparison of the microstructure as illustrated in Figures 1 and 2 against the x-ray diffraction results in Table 1 shows that there is a consistency between the microstructure and the x-ray diffraction results in that more alpha prime SiAlON phase is present in Batch No. 1924D as compared to Batch No. 1924E even through the starting powder compositions for Batch Nos. 1924 D and 1924E are the same, except for the beta silicon nitride content.
- 9. A comparison of the microstructure as illustrated in Figures 1 and 2 against the x-ray diffraction results in table 1 shows that the beta prime SiAlON grains in Batch No. 1924D achieve a greater aspect ratio (length/width) so as to be more elongate than the beta prime SiAlON grains in Batch No. 1924E even though the starting powder compositions are the same, except for the beta silicon nitride content. The more elongate beta prime SiAlON grains in Batch No. 1924D is consistent with the greater fracture toughness as compared to Batch No. 1924E.
- 10. Under my direction I had performed some experiments wherein the compositions of the starting powder mixtures and the properties of the resultant SiAlON ceramics are set forth in Table 2 below.

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Table 2 Composition and Properties of Sialon ab74012

		Comp	osition in '	Weight	Percent									,	,
	Si3N4	Si3N4	Si3N4	AIN	Al2O3	Yb2O3	β Si3N4	Hvn		Kic		α sialon	YbAIO3		Yb4SiAlO8N
•	Starck	Liha	l Iba	Starck	Caralau		Weight					A/-:			
Batch	M11	Ube SNE-03	Ube SNE-10	_	Ceralox HPA-0.5	Molycorp	Percent	GPa	st dev	MPa m1/2	st dev	Weight percent	R.I.		R.I.
1860A	43.85%			3.5			4.0	1				1			3.4
1860B	65.78%			3.5	2.6	6.2		1				j	0	4.3	6.9
1860C				3.5	2.6	6.2		16.26				ì	0	4.7	3.5
1914B			87.70%	3.5	2.6	6.2		16.53		6.45	0.19	4.	5		2.9
1914A		87.70%		3.5	2.6	6.2	c	17.62	0.34	7.08	0.23	29.:	2		3.7

Table 2 reports:

- (A) the starting compositions, selected physical properties, and selected phases present in Batch Nos. 1860A, 1860B, 1860C, 1914B and 1914A;
- (B) for each of the batches the starting powder components in weight percent wherein the components comprise Stark M11 silicon nitride powder (which has 8 weight percent beta silicon nitride), Ube SNE-10 silicon nitride powder (which has 2 weight percent beta silicon nitride), Ube SNE-03 silicon nitride powder (which has 0 weight percent beta silicon nitride), Stark Grade A aluminum nitride powder, Ceralox HPA-05 aluminum oxide powder, and Molycorp ytterbium oxide powder;
- (C) for each of the batches the beta silicon nitride content (weight percent) in the starting silicon nitride powder;
- (D) for each of the batches the Vickers hardness (18.5 kg load) reported in GPa (along with the standard deviation);
- (E) for each of the batches the fracture toughness (K _{IC}) determined by the Evans and Charles technique and reported in MPa•m^{1/2} (along with the standard deviation);

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- (F) for each of the batches the content of the alpha prime SiAlON phase in weight percent of the total of the alpha prime SiAlON phase and the beta prime SiAlON phase;
- (G) for each of the batches the relative intensity of the crystalline intergranular phase Yb₄SiAlO₈N; and
- (H) for each of the batches the relative intensity of the crystalline intergranular phase YbAlO₃.
- 11. The powder mixtures as set forth in Table 1 were processed as set out in Paragraph 6 hereof.
- 12. Figures 3 and 4 attached hereto (Exhibit B) comprise backscatter scanning electron microscope (SEM) photographs (with a 40 micrometer legend) of the polished surface wherein these photomicrographs compare the resulting microstructures of Batch Nos. 1860C and 1914A, respectively. In Figures 3 and 4 the brightest phase is the ytterbia aluminum silicon oxy-nitride grain boundary phase, the medium gray grains are the alpha prime SiAlON phase, and the darkest grains are the beta prime SiAlON phase.
- 13. A comparison of the microstructure as illustrated in Figures 3 and 4 against the x-ray diffraction results in Table 2 shows that there is a consistency between the microstructure and the diffraction results in that more alpha prime SiAlON phase is present in Batch No. 1914A as compared to Batch No. 1860C (which has no alpha prime SiAlON phase) even through the starting powder compositions for Batch Nos. 1860C and 1914A are the same, except for the beta silicon nitride content.
- 14. A comparison of the microstructure as illustrated in Figures 3 and 4 against the x-ray diffraction results show that the beta prime SiAlON grains in Batch No. 1914A achieve a greater aspect ratio (length/width) so as to be more elongate than the beta prime SiAlON grains in Batch No. 1860C even though the starting powder compositions are the same, except for the beta silicon nitride content. The more elongate beta prime SiAlON

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grains in Batch No. 1914A are consistent with the greater fracture toughness as compared to Batch No. 1860C.

15. Under my direction I had performed some experiments wherein the compositions of the starting powder mixtures and the properties of the resultant SiAlON ceramics are set forth in Table 3 below.

		Com	position in We	ight Percent								
	Si3N4	Si3N4	Si3N4	AIN Starck	Al2O3 Ceralox	Yb2O3	β Si3N4 Weight percent		st dev	Kic		α-sialon
Batch	Starck M11	Ube SNE-03	Ube SNE-10	grade A	HPA-0.5	Molycorp	veight percent	GPa		MPa		Weight percent
1417A		77.90%		7.60%	6.30%	8.20%	0	19.69	0.16			
1703B	37.96%	37.95%		8.20%	7.70%	8.20%	4.0	17.31	0.62	5.32	0.42	23.7
1605C		56.93%	18.98%	8.20%	7 70%	8 20%	0.5	10 04	0.30	6 12	0.25	د ۱

Table 3. Composition and Properties of Sialon ab153216.

Table 3 reports:

- (A) the starting compositions, selected physical properties, and selected phases present for Batch Nos. 1417A, 1703B and 1605C;
- (B) for each of the batches the starting powder components in weight percent wherein the components comprise Stark M11 silicon nitride powder (which has 8 weight percent beta silicon nitride), Ube SNE-03 silicon nitride powder (which has 0 weight percent beta silicon nitride), Ube SNE-10 silicon nitride powder (which has 2 weight percent beta silicon nitride), Stark Grade A aluminum nitride powder, Ceralox HPA-05 aluminum oxide powder, and Molycorp ytterbium oxide powder;
- (C) for each of the batches the beta silicon nitride content (weight percent) in the starting silicon nitride powder;
- (D) for each of the batches the Vickers hardness (18.5 kg load) reported in GPa (along with the standard deviation);

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- (E) for each of the batches the fracture toughness (K _{IC}) determined by the Evans and Charles technique and reported in MPa•m ½ (along with the standard deviation); and
- (F) for each of the batches the content of the alpha prime SiAlON phase in weight percent of the total of the alpha prime SiAlON phase and the beta prime SiAlON phase.
- 16. The powder mixtures as set forth in Table 3 were processed as set out in Paragraph 6 hereof.
- 17. Figures 5 and 6 attached hereto (Exhibit C) comprise backscatter scanning electron microscope (SEM) photomicrographs (with a 40 micrometer legend) of the polished surface wherein these two photomicrographs compare the resulting microstructure of Batch Nos. 1703B and 1417A, respectively. In Figures 5 and 6 the brightest phase is the ytterbia aluminum silicon oxy-nitride grain boundary phase, the medium gray grains are the alpha prime SiAlON phase, and the darkest grains are the beta prime SiAlON phase.
- 18. A comparison of the microstructure as illustrated in Figures 5 and 6 against the x-ray diffraction results in Table 3 shows that there is a consistency between the microstructure and the x-ray diffraction results in that more alpha prime SiAlON phase is present in Batch No. 1417A as compared to Batch No. 1703B even through the starting powder compositions for Batch Nos. 1417A and 1703B are the same, except for the beta silicon nitride content.
- 19. A comparison of the microstructure as illustrated in Figures 5 and 6 against the x-ray diffraction results show that the beta prime SiAlON grains in Batch No. 1417A achieve a greater aspect ratio (length/width) so as to be more elongate than the beta prime SiAlON grains in Batch No. 1703B even though the starting powder compositions are the same, except for the beta silicon nitride content.

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- 20. After reviewing the results from the experiments as reported in Tables 1, 2 and 3 and Figures 1-6, it is my opinion that restricting the beta-silicon nitride content to levels of about 1.6 weight percent or less of the starting silicon nitride powder limits the number of beta-silicon nitride particles that are nuclei for the formation of beta prime SiAlON grains during sintering so as to:
 - (A) result in the formation of fewer beta prime SiAlON grains whereby the beta prime SiAlON grains that are present develop a greater aspect ratio (length/width) (i.e., are more elongate) than if there were more beta-silicon nitride particles in the starting powder mixture;
 - (B) the presence of the elongate beta prime SiAlON grains generally improves the fracture toughness of the resultant SiAlON ceramic;
 - (C) achieve higher alpha prime SiAlON contents whereby the higher content of the alpha prime SiAlON phase incorporates more of the rare earth additives into the alpha prime SiAlON phase thereby reducing the amount of the grain boundary phase in the resultant SiAlON ceramic; and
 - (D) the higher content of alpha prime SiAlON phase generally results in a higher hardness for the resultant SiAlON ceramic.
- 21. I have reviewed the following references applied in the Office Action mailed on January 9, 2003 in the above-captioned patent application:
 - (A) English translation of Japanese Patent No. 2988966 (hereinafter JP '966);
 - (B) U.S. Patent No. 5,200,374 (hereinafter US '374);
 - (C) U.S. Patent No. 5,908,798 (hereinafter US '798);
 - (D) English translation of Japanese Laid-Open Patent Application No. 5-43333 (hereinafter JP '333); and
 - (E) U.S. Patent No. 4,574,470 (hereinafter US '470).
- 22. After a review of JP '966, it is my opinion that primarily due to the use in JP '966 of a starting silicon nitride powder that has about 7 weight percent beta silicon

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nitride (i.e., α-silicon nitride conversion of 93%), the resultant ceramic of JP '966 would not present an unexpected increase in the alpha prime SiAlON phase content or present an unexpected elongation of the beta prime SiAlON phase grains so as to achieve an unexpected increase in the hardness and the fracture toughness of the ceramic material.

23. Based upon my review of US'374, and especially the properties of the starting silicon nitride powders as set forth in Table 4 herein below, it is my opinion that

Table 4 Properties of the α -Silicon Nitride Starting Powders for US '374

Property/Powder	Col. 10	Cols. 13-14	Cols. 15-16	Col. 29
Specific Surface	17.5	10.0	11.5	11.5
Area (m ² /g)				
Particle Shape	Equiaxed	Equiaxed	Equiaxed	Equiaxed
	crystal	crystal	crystal	crystal
Formed Phase	α-phase > 95%	α -phase > 95%	α-phase > 95%	α -phase > 95%
Oxygen Content	2.3%	0.5%	1.3%	Not reported
(by weight)				
Metal Impurities	Less than 500	Less than 500	Less than 500	Less than 500
	ppm	ppm	ppm	ppm
Yeckley's	At least ≥ 2	SNE-10 (≥ 2	At least ≥ 2	At least ≥ 2
Opinion on the	wt% beta	wt% beta	wt% beta	wt% beta
Beta Content of	silicon nitride	silicon nitride)	silicon nitride	silicon nitride
the Silicon				
Nitride Powder				

in the case of the powder identified at Cols. 13-14, the α -silicon nitride starting powder is most likely SNE-10 available from Ube Industries wherein Ube SNE-10 silicon nitride powder contains about 2 or more weight percent beta-silicon nitride, and in the case of the other powders identified at Col. 10, Cols. 15-16 and Col. 29, the α -silicon nitride starting powder most likely contains at least about 2 or more weight percent beta-silicon nitride; that because of the nature of the starting powders used in US '374; namely, an α -SiAlON powder and the above-mentioned silicon nitride powders, it is my opinion that

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the resultant ceramic of US '374 would not present an unexpected increase in the alpha prime SiAlON phase content or present an unexpected elongation of the beta prime SiAlON phase grains so as to achieve an unexpected increase in the hardness and the fracture toughness of the ceramic material.

- 24. Based upon my review of US '798, the starting silicon nitride powder contains about 93 weight percent beta-silicon nitride and about 7 weight alpha-silicon nitride, and that due to the nature of the starting powder mixture, it is my opinion that the resultant ceramic of US '798 would not present an unexpected increase in the alpha prime SiAlON phase content or present an unexpected elongation of the beta prime SiAlON phase grains so as to achieve an unexpected increase in the hardness and the fracture toughness of the ceramic material.
- 25. Based upon my review of JP'333, the focus of this document is to use a very fine-grained rare earth oxide to achieve the goals of JP'333, and the silicon nitride powder is reported (in the English translation) as " α -silicon nitride powder (UBE Industries)".
- 26. The compositions set out in Table 1 hereinabove are along the lines of the composition in Example 4 of JP'333, except that the above Table 1 compositions contain about 78 percent more alumina than the alumina in Example 4 of JP'333. Based upon my experience, an increase in the alumina content in the starting powder mixture should result in a decrease in the alpha prime SiAlON content in the resultant SiAlON ceramic so that one would expect that the resultant SiAlON ceramics of Table 1 hereof would have less alpha prime SiAlON phase than Example 4 of JP'333. However, Batch No. 1924D (which has 0 beta-silicon nitride) of Table 1 hereof has almost twice as much alpha prime SiAlON phase in the resultant SiAlON ceramic as does Example 4 of JP'333. In my opinion, this difference in the alpha prime SiAlON content shows that the UBE silicon nitride powder used in JP'333 contained at least 2 weight percent (and possibly more than 2 weight percent) beta-silicon nitride.

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- 27. In light of the alpha prime SiAlON content and the nature of the starting silicon nitride powder for Examples 4 and 9 of JP'333, it is my opinion that the resultant ceramic of JP'333 would not present an unexpected increase in the alpha prime SiAlON phase content or present an unexpected elongation of the beta prime SiAlON phase grains so as to achieve an unexpected increase in the hardness and the fracture toughness of the ceramic material
- 28. Based upon my review of US'470, the starting silicon nitride powder contains about 10 weight percent beta-silicon nitride since the starting powder is described as containing 90 percent by volume of alpha silicon nitride (see Col. 5, lines 11-13 of US'470). For silicon nitride, the composition as set forth in volume percentages is approximately equivalent to the composition set forth weight percentages. In light of the nature of the starting silicon nitride powder of US'470, it is my opinion that the resultant ceramic of US '470 would not present an unexpected increase in the alpha prime SiAlON phase content or present an unexpected elongation of the beta prime SiAlON phase grains so as to achieve an unexpected increase in the hardness and the fracture toughness of the ceramic material
- 29. Kennametal makes and sells a cutting insert under the designation KYON 1540 that embodies the invention of the above-captioned patent application wherein the composition (in weight percent) of the starting powder mixture comprises: silicon nitride powder (0 % beta silicon nitride), aluminum nitride powder, alumina powder, and ytterbium oxide powder, and KYON 1540 has an alpha prime SiAlON content of between about 27 weight percent and about 33 weight percent of the total content of alpha prime SiAlON and beta prime SiAlON, a Vickers hardness (18.5 kg load) equal to between about 17.5 GPa and about 18.2 GPa, and a fracture toughness (K_{IC}) as measured by the Evans & Charles method equal to between about 6.7 MPa•m^{1/2} and about 7.2 MPa•m^{1/2}

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30. During April of 2002 Kennametal began to advertise the sale of KYON 1540 cutting inserts. Set forth below as Table 5 is a listing of what I understand to be all the advertising materials disseminated by Kennametal where some of the advertisements only advertised KYON 1540 while other advertisements advertised KYON 1540 along with other cutting inserts.

Table 5
Listing of Advertisement Materials

Exhibit	Date(s)	Description	Comments	Cost
D	April, August and September, 2002	Photographic work, printing and distribution of the Kennametal Metalcutting EXTRA Summer 2002 Catalog publication	Two pages out of a catalog comprising a 6 page cover and inside pages 3 through 74	\$54,693.88 [Total cost of preparation, printing and distribution of the entire catalog]
E	May, 2002	Preparation and printing of a "Proven in Hi Temp Alloys" Advertisement	one page (two- sided) advertisement	\$2725.67
F	September and November, 2002	Preparation and printing of a "Kyon 1540 High Performance Ceramic for Hi Temperature Alloys" Advertisement	8 page advertisement	\$9206.34
G .	January and March, 2003	Photographic work, printing and distribution of the Kennametal Metalcutting EXTRA Summer 2003 Catalog publication	Three pages out of a catalog comprising an 8 page cover and inside pages 3 through 90	\$41,159.72 [Total cost of preparation, printing and distribution of the entire catalog]

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- 31. Beginning in July 2001 (start of Kennametal's fiscal year 2002) through March 31, 2003 (the ninth month of Kennametal's fiscal year 2003) Kennametal has sold a total of 95,007 KYON 1540 cutting inserts for a total sale volume in US dollars equal to \$1,013,137.00. Attachment H hereto is a listing of the monthly sales of KYON 1540 by pieces and sales dollars.
- 32. Representative field test results are set forth in a summary of these test attached hereto as Attachments I and J. These field test results support the following general conclusions about the performance of KYON 1540:
 - (A) the tool life of the KYON 1540 cutting inserts is superior to the tool life of competitive commercial SiAlON cutting inserts wherein the typical workpiece material comprises superalloys (e.g., Inconel 718) so as to result in an improvement over available commercial SiAlON cutting inserts; and (B) in about one-half of the tests the KYON 1540 cutting inserts exhibit a tool life equal to the tool life of silicon carbide whisker-reinforced alumina cutting inserts wherein the typical workpiece material comprises superalloys (e.g., Inconel 718) so as to provide an acceptable lower cost option to silicon carbide whisker-reinforced alumina cutting inserts.
- 33. In my opinion, the performance results and conclusions about KYON 1540 were unexpected.
- 34. In my opinion the primary reason for the sale of the KYON 1540 cutting inserts is the superior performance of the KYON 1540 cutting inserts wherein this superior performance is attributable to the higher hardness (which in my opinion is due to the higher alpha prime SiAlON phase content) and the higher toughness (which in my opinion is due to the elongated beta prime SiAlON phase grains) of the KYON 1540 ceramic material.
- 35. Referring to the nature of the sales of KYON 1540, some of these sales comprise replacements of previous sales of Kennametal KYON 2000 (an α - β SiAlON

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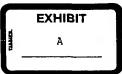
[no Yb] that has between about 20 weight percent and about 40 weight percent α-SiAlON and uses a starting silicon nitride powder that contains between about 6 and about 7 weight percent beta silicon nitride) ceramic cutting inserts and Kennametal KYON 2100 (β-SiAlON) ceramic cutting inserts, new sales due to the superior performance of KYON 1540 against competitor's SiAlON ceramic cutting inserts, and new sales due to the equivalent performance of KYON 1540 against silicon carbide whisker reinforced alumina ceramic cutting inserts.

DECLARANT SAYS NOTHING FURTHER

All statements made of my own knowledge are true and all statements made on information and belief are believed to be true. I have been warned that willful false statements and the like are punishable by fine or imprisonment, or both (18 USC1001) and may jeopardize the validity of this application or any patent issuing thereon.

Date 5 June 03

Kennametal1600/K1688Declaration(rev)Yeckley-002



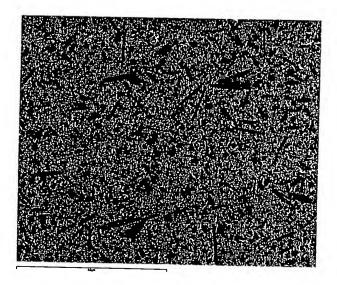


Figure 1. Backscatter SEM image of Sialon composition ab83612 Batch 1924E microstructure.

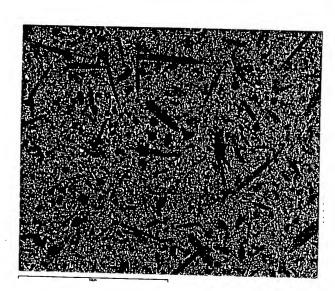


Figure 2. Backscatter SEM Image of Sialon composition ab83612 Batch 1924D microstructure

EXHIBIT
B
B

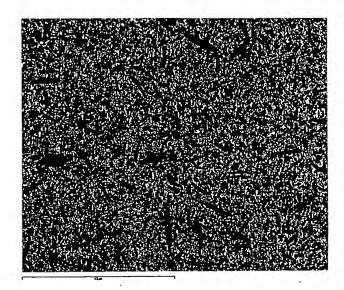


Figure 3. Backscatter SEM Image of Slalon composition ab74012 Batch 1860C microstructure

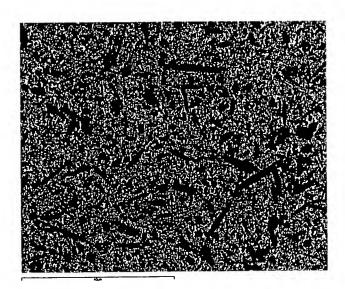


Figure 4. Backscatter SEM image of Sialon composition ab74012 Batch 1914A microstructure

EXHIBIT

C

C

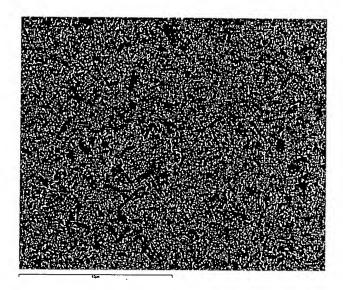


Figure 5. Backscatter SEM image of Slalon composition ab153216 Batch 1703B microstructure

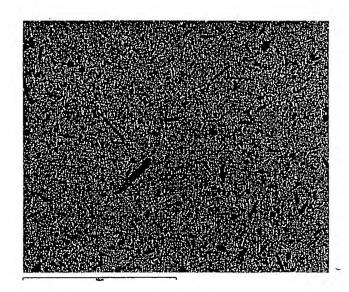


Figure 6. Backscatter SEM Image of Slalon composition ab153216 Batch 1417A microstructure

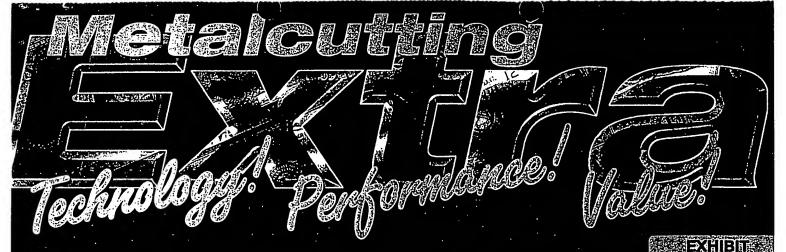


Kyon 1540 Field Test Summary

	oy Grad		Insert	Speed	DOC in.	٠. ١	Comments
IN7		-	RNG45	1200		0.0	
<u></u>	Ky20		RNG45	850	0.120	0.00	
IN70			RNG45	1,100	0.120	0.01	
	Ky20		RNG45	800	0.100	0.01	
IN71				800	0.15	0.0	
	SX5		1	800	0.15	0.0	The second twice 3V2
IN71				1000	0.125	0.00	
	· CC67	0		1000	0.125	0.00	- Tivi 1040 Equialent in C.C.n/n
IN71	8 KY154	10		800	0.125	0.00	···
<u></u>	CC67	0	1	800	0.125	0.00	
IN71	8 KY154	0		650	.10012	0.00	f _
	Ky210	0		650	.10012	0.00	- 1.0% 1.00dcttvtty tillblovettlett
IN71	8 KY154	0		271	0.15		
l	Ky200	o l	ļ	2''	0.15	0.0	1 000 bo of Mo 14 Older ellered
1	Ky210		I			1	estimate 5000-6000 per year.
IN706			SNG644		0.000	4	
	KY154			800	0.265	0.00	The follows chippling
	134	~	SNG644			1	In another application with 15 degree lead
IN718	KY1540	. 					angle, KY1540 moderate DOCN
114/18	WG300		SNG644	706	.150350	0.000	KY1540 ran fairly equal to WG-300
IN718			SNGN644			L	slightly more flank wear, but less DOC notching
1147 10	1		SNG654	700	0.15	0.005	Equivalent to WG300
111740	WG300		SNG654				
IN718			1	900	.0575	0.008	KY1540 ran equal to Q8, Not to End of life.
	KY1540	' j	RNG45	1 1		l	The state of the s
·	Q8		RNG45T0320			1	
Incone		1	RNG45T0420	800	0.05	0.006	KY1540 no flaking or chienian On flaki
D979	Q8	1	RNG45T0320				KY1540 no flaking or chipping, Q8 flaking and
		<u> </u>		1 1			chipping. Operator sometimes has to make
IN718	KY1540		RPGV45T0420	900	0.05	0.005	second pass with Q8 due to chipping.
_	WG300	l	RPGN4VT1		0.00	0.000	KY1540 had Slightly more edge wear, but
IN718	KY1540		RPGV35T0420	900	.01502	0.005	completed cut.
	WG300	ļ	RPGV35T0320	300	.0,1002	0.005	Equivalent to WG300
	H13A			 			
IN901	Carbide		CNMG644QM	100	0.15	0.040	KY1540 implemented, estimated annual savings
	KY1540	ŀ	RNG45T0420	1000	0.15	0.016 0.01	\$80,000
			1	1000	0.10	0.04	
IN713	Carbide	38 Rc	CNMG431	100	0.005		KY1540 implemented, estimated annual savings
	KY1540	00110	CNGA433T0420	100 600	0.025	0.006	\$60,000
	1		ONGA43310420	000	0.1	0.006	
IN718	Carbide	43 Rc	Character				KY1540 implemented, estimated annual savings
111/10	KY1540	43 KC	SNMA644	73	0.05	0.015	\$3;000
IN718	SX5		RNG45T0420	271	0.05	0.01	hanner and the second s
111/10	KY1540	44 Rc	RNG65T RNG65T0420	600	0.25	0.006	KY1540 2X life, estimated annual savings \$1,500
181740		77 I\U	A STATE OF THE PARTY OF THE PAR	600	0.25	0.006	
IN718	WG300 KY1540		SNG654	500		0.009	Equivalent to WG300, less notching on KY1540
IN718			SNG645	500	0.13	0.009	Estimated annual savings \$2,000
111/10	Q8 KY1540	39 Rc	RNG45T00320	800		0.007	KY1540 performance is equivalent with less wear,
	171 1040	35 KC	RNG45T03420	800	0.15	0.007	estimated annual savings \$3000
IN718	1,4/0000		DD0.40		T		KY1540 equivalent to WG300, estimated annual
111/10	WG300 KY1540	30 Pa	RPGV35T0320	900		0.005	savings \$3,500
11:40	171 1340	39 Rc	RPGV45T0420	900	0.02	0.005	
ellite		58 Rc			T		KY1540 insert life is 4X WG300, speed is limited to
ay in	WG300	Inlay	CNGA433T1A	600	0.125	0.004	600 sfm max, estimated annual savings \$2,200
ا		28 Rc		1	İ		
	KY1540	Steel	CNGA433S0420	600	0.125	0.004	
	WG300		RPGN3VT1	600	0.075	.0024	KY1540 broke, did not complete cut
	KY1540	- 1	RPGV35T0320			:0024	The mor complete out

KY1540 TPR Summary

							And the state of t
Alloy	Grade	Hardnes	s Insert	Speed sfm	DOC in.	feed ipr	Comments
IN706	KY1540		RNG45	1200	0.12	0.012	Replacing Ky2000 and Ky2100
	Ky2000		RNG45	850	0.120	0.0080	with KY1540
IN706	KY1540 Ky2000		RNG45	1,100	0.120	0.0145	l ·
IN706	Ky2100		RNG45	800	0.100	0.0100	1001-101
114700	KY1540		SNG644 SNG644	800	0.265	0.006	KY1540 less chipping In another application with 15 degree lead angle, KY1540 moderate DOCN
IN706	KY2000 KY1540		SNG656 SNG657	800 800	0.2 0.2	0:008 0.008	KY1540 5X iife, estimated annual savings \$750
IN718	KY1540			800	0.15	0.007	KY1540 life twice SX5
ļ	SX5	1		800	0.15	0.007	Entered blanket order
IN718	KY1540 CC670			1000	0.125	0.006	KY1540 equialent to CC670
IN718	KY1540	-		1000 800	0.125 0.125	0.006 0.006	
1147 10	CC670			800	0.125	0.006	
IN718	KY1540			650	.100120	0.012	70% Productivity improvement
	Ky2100		e	650	.100120	0.007	
IN718	KY1540			271	0.15	0.01	300 pc SPM014 order entered
	Ky2000				-3		estimate 5000-6000 per year.
IN718	Ky2100 KY1540	↓	SNG644	706	.150350	0.006	KY1540 ran fairly equal to WG-300
111/16	WG300		SNGN644	700	. 150550	0.006	slightly more flank wear, but less DOC notching.
IN718	KY1540	 	SNG654	700	0.15	0.005	Equivalent to WG300
"""	WG300		SNG654	1	0.10	0.000	Equivalent to vvocoo
IN718	KY1540	39 Rc	RNG45T0420	900	.0575	0.008	KY1540 ran equal to Q8, Not to End of life.
1	KY1540		RNG45				
	Q8		RNG45T0320		<u> </u>		
IN718	KY1540		RPGV45T0420	900	0.05	0.005	KY1540 had Slightly more edge wear, but
IN718	WG300	 	RPGN4VT1 RPGV35T0420	900	.01502	0.005	completed cut. Equivalent to WG300
1111/10	KY1540 WG300		RPGV35T0420	4	.01502	0.003	Equivalent to vvG300
	110000		0 7 00 10020			-	KY1540 implemented, estimated annual savings
IN718	Carbide	43 Rc	SNMA644	73	0.05	0.015	\$3,000
	KY1540		RNG45T0420	271	0.05	0.01	
		l					
IN718	SX5 KY1540	44 Rc	RNG65T RNG65T0420	600 600	0.25 0.25	0.006 0.006	KY1540 2X life, estimated annual savings \$1,500
IN718	WG300	44 RC	SNG654	500	0.23	0.009	Equivalent to WG300, less notching on KY1540
114710	KY1540		SNG645	500	0.13	0.009	Estimated annual savings \$2,000
							
IN718	Q8		RNG45T00320	800	0.15	0.007	KY1540 performance is equivalent with less wear,
<u> </u>	KY1540	39 Rc	RNG45T03420	800	0.15	0.007	estimated annual savings \$3000
111740	WC200	i .	DDC) (SETOSO	000		0.005	KY1540 equivalent to WG300, estimated annual
IN718	WG300		RPGV35T0320	900	0.02	0.005	savings \$3,500
	KY1540	39 Rc	RPGV45T0420	900	0.02	0.005	·
IN718	WG300		RPGN3VT1	600	0.075	0.0024	KY1540 broke, did not complete cut
					İ		
	KY1540		RPGV35T0320	600	0.075	0.0024	WG300 completed cut
IN713	Carbide	38 Rc	CNMG431	100	0.025	0.006	KY1540 implemented, estimated annual savings
1147 15	Carbiac	30110	CNGA433T042	100	0.023	0.000	\$60,000
	KY1540		· 0	600	0.1	0.006	
	H13A						KY1540 implemented, estimated annual savings
IN901	Carbide KY1540		CNMG644QM	100	0.15	0.016	\$80,000
Inconel	KY1540 KY1540		RNG45T0420 RNG45T0420	1000 800	0.15	0.01 0.006	IV/4E 40 no fielder or objector OC fielder and
D979	Q8		RNG45T0420	800	0.05	0.006	KY1540 no flaking or chipping, Q8 flaking and chipping. Operator sometimes has to make
						2.000	second pass with Q8 due to chipping.
							, , , , , , , , , , , , , , , , , , , ,
Stellite							KY1540 insert life is 4X WG300, speed is limited to
Infay in	WG300	58 Rc Inlay	CNGA433T1A	600	0.125	0.004	600 sfm max, estimated annual savings \$2,200
Alloy Steel	KY1540	28 Rc Steel	CNGA433S042 0	600	0.125	0.004	
Waspaloy		40Rc	SNG644	625	0.125		VV2100 2V life of KV1540, phinning
• •aspaioy	KY1540	701/6	SNG644 SNG644	625	0.35	0.007	KY2100 3X life of KY1540, chipping
	H13A		5.100-74				KY1540 20X life at higher speed, estimated annual
A286	Carbide	33 Rc	CNMG644QM	140	0.15	0.016	savings of \$11,000
	KY1540.	·	RNG45	1000	0.15	0.01	
		46 Rc	RNG45T1	600	0.02		WG300 1.5x life of KY1540
	KY2000 KY1540		RNG45 RNG45	600 600	0.02 0.02	0.01 0.01	KY1540 1.5x times life of KY2000
	1711040 1		1111040	000	0.02	0.011	



Summer 2002



PROVEN SOLUTIONS

to boost your machining productivity and profitability.

FEATURING:

New A4 Groove & Turn tooling!

New KC9110 & KC9125 turning grades of steel machining!

New NGE-B indexable insert end mills and shell mills!

New K284 & K285 solid carbide drills for high-temperature alloys!

New high-temp alloy milling system with Kyon 2100 ceramic inserts!

New Shrink-Fit induction heating system and toolholders!

Kennametal.

The Metalcutting Authority."



PROVEN. IN HIGH-TEMP MACHINING.

Y1540 is our newest and iost advanced SiALON ceramic iaterial for turning highimperature alloys. nhancements in both its ardness and toughness levels nable this grade to run at igher feed rates than KY2100, hile still maintaining utstanding depth-of-cut notch sistance. Result: higher roductivity with much longer ol life on these difficult to achine materials.

ne two edge prep options ione or .004" K-land) enable erformance optimization over wide range of cutting oplications, from roughing rough finishing cuts.

application Range

ickel-Base Alloys:

50 sfm - 1000 sfm .003" - 0.012" feed rates

obalt-Base Alloys:

00 sfm - 900 sfm .003" - 0.012" feed rates

on-Base Alloys:

00 sfm - 800 sfm .003" - 0.016" feed rates

> Need technical assistance? Call our



Tech Hotline! 800/835-3668

TURNING

Introducing **Kyon 1540 Inserts**



medium machining

ANSI CNG433 CNG434 CNG432T0420 CNG433T0420 CNG434T0420 CNG544T0420 RNG43 RNG45 RNG43T0420 RNG45T0420 SNG644 SNG656

SNG432T0420 SNG433T0420 SNG434T0420 SNG452T0420 SNG453T0420 SNG454T0420 SNG644T0420 TNG433T0420 TNG434T0420 TNG453T0420 TNG454T0420



medium machining

ANSI

CNGA432 **CNGA433** CNGA434 CNGA432T0420 CNGA433T0420 CNGA434T0420 CNGA643T0420 DNGA544 DNGA432T0420 DNGA433T0420 SNGA433T0420 SNGA434T0420 TNGA432T0420 TNGA433T0420



medium machining

ANSI

RPGV35 RPGV45 RPGV35T0420 RPGV45T0420

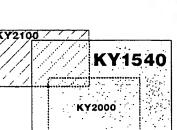


medium machining

ANSI

RCGV23 RCGV35 RCGV45

RCGV35T0420 RCGV45T0420



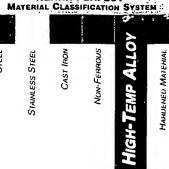
higher feed/toughness

higher speed/wear resistance

RPG medium machining ANSI

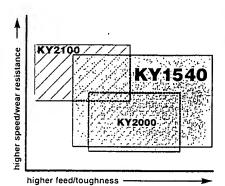
RPG32 RPG43

> Kennamejakobiline opros KYON, SIALON CETAMIC CITAGE Indicates uncoated grades Suited for material group high-temp alloys





Kyon 2100 Inserts for High Speed Machining **Nickel-Base Alloys**





TOOLtips

New process and material technologies derived from grade KY 1540 are also being applied to our existing KY2100 SIALON ceramic inserts which remain the preferred choice for ... medium machining nickel-base alloys such as Inconel, Hastelloy, and Waspaloy at high speeds and light to medium feeds.

As well as high thermal and mechanical shock resistance, Kyon 2100 has outstanding resistance to depth-of-cut notching, the main failure mode when machining high-temp alloys.



medium machining

ANSI

RPGV-35 **RPGV-35T 0420 RPGV-45T 0420**



medium KY2100 machining

NG & T

ANSI CNG-432T 0820 CNG-433T 0420 CNG-434T 0420 RNG-43 RNG-45 RNG-43T 0420 RNG-45T 0420 SNG-433 SNG-434 SNG-644 SNG-434T 0420 SNG-453T 0420

medium machining

ANSI CNGA433 CNGA-433T 0420 CNGA-434T 0420

SNGA-433T 0420

KY2100 machining ANSI

RCGV-23 RCGV-35 RCGV-45 RCGV-35T 0420 RCGV-45T 0420

Enroll in our five-day application engineering course!



(724) 539-6828

PROVENTINE ALLOYS.

NEW!

Kyone 1540 ...

A HIGH-PERFORMANCE CERAMIC SPECIFICALLY ENGINEERED FOR HIGH-TEMPERATURE ALLOYS.

Key Benefits

A Proven Performer...

- machining a broad range of high-temperature alloys – including Inconels, Waspaloys, and Hastelloys
- involving a wide variety of machining conditions, including cuts with interruptions and scale
- as a cost-effective alternative to whisker ceramic materials

This new Kennametal grade is your first choice Kyon ceramic solution specifically designed for a wide variety of high-temperature alloy machining conditions.

Key Features

- edge prep... offered with a hone or .004 inch
 (0,1 mm) K-land
- the two edge prep options enable performance optimization over a wide range of cutting applications, from roughing through finishing cuts

A patented composition enables Kyon 1540 to provide improved hardness (wear) and toughness (strength) properties over conventional sialon ceramics.



Kennametal.

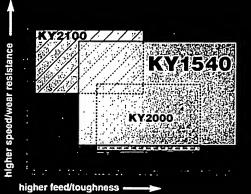
The Metalcutting Authority".

www.kennametal.com

COURT by Kennametal Inc., Labour, Mr. - All (balto (specyed. - All-opto)DR

New Kyono 1540...

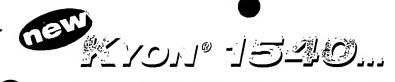
Kennametal Ceramic Grades for Machining High-Temperature Alloys



Kyon® 1540 Insert OfferingStandard inserts are available June 1, 2002.

insert catalog number	ISO catalog number
CNG432T0420	CNGN120408T01020
CNG433	CNGN120412E
CNG433T0420	CNGN120412T01020
CNG434	CNGN120416E
CNG434T0420	CNGN120416T01020
CNG544T0420	CNGN160616T01020
CNGA432	CNGA120408E
CNGA432T0420	CNGA120408T01020
CNGA433	CNGA120412E
CNGA433T0420	CNGA120412T01020
CNGA434	CNGA120416E
CNGA434T0420	CNGA120416T01020
CNGA643T0420	CNGA190612T01020
DNGA432T0420	DNGA150408T01020
DNGA433T0420	DNGA150412T01020
DNGA544	DNGA190616E
KGF81872	KGF25047608E
KGF82504	KGF25063516E
KGF83752	KGF25095208E
KGF92504	KGF28063516E
RCGV23	RCGX060400E
RCGV35	RCGX090700E
RCGV35T0420	RCGX090700T01020
RCGV45	RCGX120700E
RCGV45T0420	RCGX120700T01020
RNG43	RNGN120400E
RNG43T0420	RNGN120400T01020

insert catalog number	ISO catalog number				
RNG45	RNGN120700E				
RNG45T0420	RNGN120700T01020				
RPG32	RPGN090300E				
RPG43	RPGN120400E				
RPGV35	RPGX090700E				
RPGV35T0420	RPGX090700T01020				
RPGV45	RPGX120700E				
RPGV45T0420	RPGX120700T01020				
SNG432T0420	SNGN120408T01020				
SNG433T0420	SNGN120412T01020				
SNG434T0420	SNGN120416T01020				
SNG452T0420	SNGN120708T01020				
SNG453T0420	SNGN120712T01020				
SNG454T0420	SNGN120716T01020				
SNG644	SNGN190616E				
SNG644T0420	SNGN190616T01020				
SNG656	SNGN190724E				
SNGA433T0420	SNGA120412T01020				
SNGA434T0420	SNGA120416T01020				
TNG433T0420	TNGN220412T01020				
TNG434T0420	TNGN220416T01020				
TNG453T0420	TNGN220712T01020				
TNG454T0420	TNGN220716T01020				
TNGA432T0420	TNGA220408T01020				
TNGA433T0420	TNGA220412T01020				
TPG322	TPGN160308E				
Customer Service (USA ar	nd Canada)				



د التنابات تعطی المحتود التنابات المحتود

versus Whisker ceramic

higher productivity versus carbide tools

Cost offective

High-Performance Ceramic for High-Temperature Alloys







Lathe



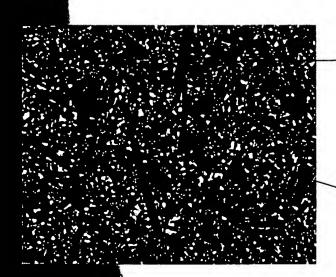
Introducing New Kyon 1540...

your first choice ceramic solution specifically designed for a wide variety of high-temperature alloy machining conditions!

Kyon 1540 is a Proven Performer...

- · in turning and milling applications
- machining a broad range of materials, especially Inconels and other nickel-base hightemperature alloys
- involving a wide variety of machining conditions, including cuts with interruptions and scale
- as a cost-effective alternative to whisker ceramic materials





whisker-shaped beta sialon grains enhance fracture toughness

uniform alpha sialon grain size and composition enhance hardness

grade	coating	Insert styles	composition and application	C class	ISO class
KY1540		turning boring profiling milling	composition: KY1540 is the latest and most advanced sialon material ever developed. application: Combines excellent wear properties, fracture toughness, and thermal shock resistance for general purpose to finish machining of high-temperature alloys. Provides superior depth of cut notch resistance as compared to whisker ceramics.	C4	M10-M25 K05-K15

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PROVEN. In Aerospace and Power Generation.

Our unique combination of properties found in Kyon 1540 significantly broaden the effective machining range of sialon ceramics in hightemperature alloy applications!

- runs at higher feeds and speeds for increased productivity
- · performs at lower speeds equally well, when restricted by equipment or fixturing limitations, for improved tool life

1st Step - Select the Insert Geometry

▼▼ Medium Machining





..NG-T

R.GV-T

Finishing





..GA

R.GV

Calaat tha Coada

2nd Step – Select the Grade									
		Insert Geometry							
Cutting Condition	•	R.GV	GA	NG-T	R.GV-T				
		***	777	**	**				
heavily interrupted cut	6								
lightly interrupted cut	®	KY1540	* KY1540.	ind)	. 1564				
varying depth of cut, casting or forging skin	69	KY1540	≠ KY1540 👯	1000	e to sup				
smooth cut, pre-turned surface	(A)	KY2100	KY2100	er say	211 pm				

.0025

0,063

.160 .100

.060

.016 .010

.004

.0021

depth-of-cut

(inch) .025

0,16

.010

feed rate (in/rev)

.016

.006

.025

KENNAME

KY4540

KY2000

0.63 (mm)

0,25

0,10 0,063

Starting Conditions

m/min

.060

.040

3rd Step - Select the Cutting Speed

	Speed - sfm (m/min)							/min)				
Grade	50 (15)	150 (45)	250 (75)	350 (110)	450 (140)	550 (170)	650 (200)	750 (230)	850 (260)	950 (290)	1050 (320)	
KY1540	300000						©		②			



Kenloc &	insert	ISO	IC		т		R		В		н	
Kendex	catalog number	catalog number	inch	mm	inch	mm	inch	mm	inch	mm	inch	mm
CNGA / CNGA-T	CNGA-432 CNGA-433 CNGA-434	CNGA 12 04 08E CNGA 12 04 12E CNGA 12 04 16E	1/2 1/2 1/2	12,70 12,70 12,70	3/16 3/16 3/16	4,76 4,76 4,76	1/32 3/64 1/16	0,8 1,2 1,6	.1216 .1129 .1042	3,088 2,867 2,647	.203 .203 .203	5,16 5,16 5,16
B 80°	with T-land (.004" x 20") CNGA-432T-0420. CNGA-433T-0420. CNGA-434T-0420. CNGA-643T-0420.	CNGA 12 04 08T 01020 CNGA 12 04 12T 01020 CNGA 12 04 16T 01020 CNGA 19 06 12T 01020	1/2 1/2 1/2 1/2 3/4	12,70 12,70 12,70 12,70	3/16 3/16 3/16 1/4	4,76 4,76 4,76 6,35	1/32 3/64 1/16 3/64	0,8 1,2 1,6	.1216 .1129 .1042 .1823	3,088 2,867 2,647 4,632	.203 .203 .203 .312	5,16 5,16 5,16 7,93
DNGA / DNGA-T	DNGA-544	DNGA 19 06 16E	5/8	15,88	1/4	6,35	1/16	1,6	.2914	7,402	.250	6,35
B 55°	with T-land (.004" x 20°) DNGA 432T 0420 DNGA 433T 0420	DNGA 15 04 08T 01020 DNGA 15 04 12T 01020	1/2 1/2	12,70 12,70	3/16 3/16	4,76 4,76	1/32 3/64	0,8 1,2	.2550 .2368	6,477 6,014	.203 .203	5,16 5,16
SNGA-T	with T-land (.004" x 20°) SNGA-433T 0420 SNGA-434T 0420	SNGA 12 04 12T 01020 SNGA 12 04 16T 01020	1/2 1/2	12,70 12,70	3/16 3/16	4,76 4,76	3/64 1/16	1,2 1,6	.0841 .0777	2,137 1,973	.203 .203	5,16 5,16
TNGA-T	with T-land (.004" x 20°) TNGA-432T 0420 TNGA-433T 0420	TNGA 22 04 08T 01020 TNGA 22 04 12T 01020	1/2 1/2	12,70 12,70	3/16 3/16	4,76 4,76	1/32 3/64	0,8	.7188 .7031	18,256 17,859	.203 .203	5,16 5,16
CNG / CNG-T	CNG-433 CNG-434	CNGN 12 04 12E CNGN 12 04 16E	1/2	12,70 12,70	3/16 3/16	4,76 4,76	3/64 1/16	1,2 1,6	.1129 .1042	2,867 2,647	_	
R B 80°	with T-land (.004" x 20°) CNG-432T 0420 CNG-434T 0420 CNG-544T 0420	CNGN 12 04 08T 01020 CNGN 12 04 12T 01020 CNGN 12 04 16T 01020 CNGN 16 06 16T 01020	1/2 1/2 1/2 1/2 5/8	12,70 12,70 12,70 15,88	3/16 3/16 3/16 1/4	4,76 4,76 4,76 4,76 6,35	1/32 3/64 1/16	0,8 1,2 1,6 1,6	.1216 .1129 .1042	3,088 2,867 2,647 3,529	- - -	- - - -
RNG / RNG-T	RNG-43 RNG-45	RNGN 12 04 00E RNGN 12 07 00E	1/2 1/2	12,70 12,70	3/16 5/16	4,76 7,94	_		=	-		
IC T	RNG-65 with T-land (.004" x 20°) RNG-43T 0420 RNG-45T 0420 RNG-65T 0420	RNGN 19 07 00E RNGN 12 04 00T 01020 RNGN 12 07 00T 01020 RNGN 19 07 00T 01020	1/2 1/2 1/2 3/4	19,05 12,70 12,70 19,05	3/16 5/16 5/16	4,76 7,94 7,94	- - -	- - -	- - -	- - -		- - -
RPG	RPG-32	RPGN 09 03 00E RPGN 12 04 00E	3/8 1/2	9,53 12,70	1/8 3/16	3,18 4,76	=	<u>-</u>	+=	+=-	=	+=
, ic iii	RPG-43	AFGN 12 04 00E	1/2	12,70	5,10	1,70						



Kendex &	insert	ISO	1	С		т		R	1	В
Positive Profiling	catalog number	catalog number	inch	mm	inch	mm	inch	mm	inch	mm
SNG / SNG-T	SNG-644 SNG-656	SNGN 19 06 16E SNGN 19 07 24E	3/4 3/4	19,05 19,05	1/4 5/16	6,35 7,94	1/16 3/32	1,6 2,4	.1294 .1165	3,288 2,959
B 90°	with T-land (.004" x 20°) SNG-432T-0420	SNGN 19 07 24E	1/2	12,70	3/16	4,76	1/32	0,8	.0906	2,301
A HTH	SNG-433T,0420 SNG-434T,0420 SNG-452T,0420 SNG-453T,0420 SNG-454T,0420	SNGN 12 04 12T 01020 SNGN 12 04 16T 01020 SNGN 12 07 08T 01020 SNGN 12 07 12T 01020 SNGN 12 07 16T 01020	1/2 1/2 1/2 1/2 1/2	12,70 12,70 12,70 12,70 12,70 12,70	3/16 3/16 5/16 5/16 5/16	4,76 4,76 7,94 7,94 7,94	3/64 1/16 1/32 3/64 1/16	1,2 1,6 0,8 1,2 1,6	.0841 .0777 .0906 .0841 .0777	2,137 1,973 2,301 2,137 1,973
	SNG-644T 0420	SNGN 19 06 16T 01020	3/4	19,05	1/4	6,35	1/16	1,6	.1294	3,288
TNG-T	with T-land (.004" x 20")									
60% B C T T	TNG-433T 0420 TNG-434T 0420 TNG-453T 0420 TNG-454T 0420	TNGN 22 04 12T 01020 TNGN 22 04 16T 01020 TNGN 22 07 12T 01020 TNGN 22 07 16T 01020	1/2 1/2 1/2 1/2 1/2	12,70 12,70 12,70 12,70	3/16 3/16 5/16 5/16	4,76 4,76 7,94 7,94	3/64 1/16 3/64 1/16	1,2 1,6 1,2 1,6	.7031 .6875 .7031 .6875	17,859 17,463 17,859 17,463
TPG B IC B T T T T T T T T T T T T	TPG-322	TPGN 16 03 08E	3/8	9,53	1/8	3,18	1/32	0,8	.5313	13,494
RCGV / RCGV-T	RCGV-23	RCGX 06 04 00E	1/4	6,35	.188	4,78		-		
230°1−	RCGV-35	RCGX 09 07 00E	3/8	9,53	.312	7,92				
1C 7°	RCGV-45 with T-land (.004" x 20°) RCGV-35T 0420	RCGX 12 07 00E RCGX 09 07 00T 01020	3/8	12,70 9,53	.312	7,92			_	
- T -	RCGV-45T-0420	RCGX 12 07 00T 01020	1/2	12,70	.312	7,92	-	_	_	_
RPGV / RPGV-T	RPGV-35	RPGX 09 07 00E	3/8	9,53	.312	7,92	_	-	_	-
λ30°	RPGV-45	RPGX 12 07 00E	1/2	12,70	.312	7,92	-		_	
110	with T-land (.004" x 20°) RPGV-35T 0420 RPGV-45T 0420	RPGX 09 07 00T 01020 RPGX 12 07 00T 01020	3/8	9,53 12,70	.312 .312	7,92 7,92		-	<u>-</u>	
										

Kendex V-Bottom Deep Grooving System	insert catalog number	ISO catalog number	W Inch mm		R inch mm		L inch mm		T inch mm		B inch mm	
KGF ·	KGF-8187-2	KGF 25 04 76 08E	.187	4,32	.031	0,79	1	25,40	.328	8,31	-	
, B7,	KGF-8250-4	KGF 25 06 35 16E	.250	6,35	.062	1,58	1	25,40	.328	8,31	-	_
5,	KGF-8375-2	KGF 25 09 52 08E	.375	9,52	.031	0,79	1	25,40	.328	8,31	_	_
2° W T T T	KGF-9250-4	KGF 28 06 35 16E	.250	6,35	.062	1,58	1 1/8	28,58	.328	8,31	-	-

METALCUTTING

TECHNOLOGY

DVERT

PERFORMANCE

VALUE

Spring 2003

PROVEN SOLUTIONS

to boost your machining productivity and profitability!

FEATURING:

New KC915M milling grade for cast and ductile iron!

New KC935M milling grade for steel, stainless steel, and ductile iron!

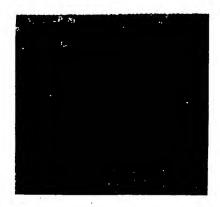
New KD1405 diamond sheet turning and milling grade for aluminum!

New Kyon 1540 ceramic inserts for high-temp/heat-resistant alloys!

New KB9640, the industry's first CVD alumina-coated PCBN grade!

See These Products
at

March 24-27,
March 24-27,
LA Convention Center,
Booth #3032



TOOLtips

Our new KDNR and KIPR Cutter Bodies are designed specifically for Kyon 2100 grade ceramic inserts and will help you attain surface speeds in excess of 3,000 sfm. The cutter's rugged clamping system was developed to operate at elevated spindle speeds while machining nickelbased, heat-resistant alloys such as Inconel 625, 718, and other difficult-to-machine materials. These cutters, with Kyon 2100 inserts, generally work best when run dry and supplied with a through-the-spindle air blast for chip evacuation.

Grade KY2100 is a silicon nitride (sialon) based ceramic. Its high thermal and mechanical shock resistance makes it ideal for light to heavy rough milling of PH and 300 series stainless steel as well as nickel- and cobalt-base heat-resistant materials such as Inconel and Waspaloy. Kyon 2100 has outstanding resistance to depth-of-cut notching, the main failure mechanism when machining high-temp alloys.

Kyon 1540 is our newest and most advanced sialon grade. It has moderately higher thermal and mechanical shock resistance than KY2100. Grade KY1540 should be used for low to high-speed rough milling of nickel, cobalt, and iron-base heat-resistant alloys at speeds ranging from 2000 to over 3000 sfm.

PROVEN. IN HIGH-TEMP MACHINING.



New Milling System with Kyon 2100 Ceramic Inserts for High-Temp/ Heat-Resistant Alloys

Increase Surface Speeds to Over 3,000 SFM and Reduce Cut Time by 60% or More!



Kennametal Engineer:

Doug Armond and Bob Fisher

Market: Oil and Gas

Product: Rough mill flats on 6 1/4" O.D. tube

Material: Inconel 718

	KENNAMETAL	COMPETITION
cutter	KDNR300RN40C4	indexable insert cutter
grade	KY2100	coated carbide
insert	RNG45T0420	octogon
speed	3,200 sim	80 slm
eed	.003 ipt (61 ipm)	.007 ipt (3,7 ipm)
axial d.o.c	.080	.100
adial d.o.c	1.50	1.50



Kennametal Engineer: Ed Skrzynski

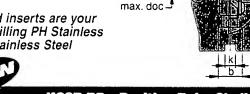
Market: Aerospace Product: Jet engine flange

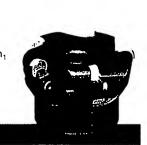
Material: AMS 5772 - Haynes 188 26 HRC

	KENNAMETAL	COMPETITION
cutter	KDNR400RN40C5	indexable insert cutter
grade	KY2100	coated carbide
insert	RNG45T0420	OFER070405TNME10
speed	3,500 sfm	52 sfm
feed	.004 ipt (92 ipm)	.0035 ipt (1 ipm)
axial d.o.c	.035	.035
radial d.o.c	2.50	2.50

cos navings. The cost savings was septemented of one of the cost savings was \$250,000 annually.

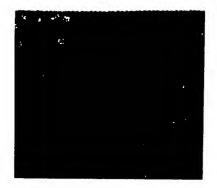
These cutters and inserts are your best choice for milling PH Stainless and 300 Series Stainless Steel





KSSR-RP - Positive Rake Shell Mills

cutting diameter d ₁	catalog number	insert catalog number	number of inserts	height h;	:	bore b	:	keyway k	clamp	:	clamp screw		max. rpm		weight lbs.
2.00 .	KSSR200RP430C3	: RPG43	3	2.00	:	.75	Ť	.331	KCI-3	_	S-1997	÷	16,000	-	<2.0
2.00	KSSR200RP430F3	RPG43	4	2.00	'	.75	i	.331	KCI-3		S-1997	!	16,000		₹.0
2.50	KSSR250RP430C3	RPG43	4	2.00	į	.75	1	.331	KCI-3	i	S-1997	i	14,500	:	₹2.0
3.00	KSSR300RP430C4	RPG43	5	2.00		1.00		.390	KCI-3	:	S-1997	:	13,500		2.0
4.00	KSSR400RP430C5	RPG43	6	2.00	;	1.25	:	.515	KCI-3	!	S-1997	ł	11,500		3.5



PROVEN. IN HIGH-TEMP MACHINING.

TOOLtips

KY1540 is our newest and most advanced sialon ceramic material for turning hightemperature alloys. Enhancements in both its hardness and toughness levels enable this grade to run at higher feed rates than KY2100, while still maintaining outstanding depth-of-cut notch resistance. Result: higher productivity with much longer tool life on these difficult to machine materials.

The two edge prep options (hone or .004 K-land) enable performance optimization over a wide range of cutting applications, from roughing through finishing cuts.

Application Range

Nickel-Base Alloys:

450 sfm - 1000 sfm .003 - .012 feed rates

Cobalt-Base Alloys:

300 sfm - 900 sfm .003 - .012 feed rates

Iron-Base Alloys:

300 sfm - 800 sfm .003 - .016 feed rates

> Need technical assistance? Call our Tech Hotline!



800/835-3668

TURNING

Introducing **Kyon 1540 Inserts**



medium machining	540
	. : <u>=</u>
ANSI	⊊
CNG433	1 •
CNG434	. •
CNG432T0420	. •
CNG433T0420	' ♦
CNG434T0420	' ♦
CNG544T0420	, ♦
RNG43	•
RNG45	. •
RNG43T0420	: ♦
RNG45T0420	•
SNG644	• · · · · ·
SNG656	. •
SNG432T0420	↑ •
SNG433T0420	•
SNG434T0420	•
SNG452T0420	* •
SNG453T0420	•
SNG454T0420	. •
SNG644T0420	•
TNG433T0420	•
TNG434T0420	•

TNGASSTOASO

	NG453104 NG454T04	
er speed/wear resistance	KY21	KY1540 KY2000

higher feed/toughness



medium machining
ANSI
CNGA432
CNGA433
CNGA434
CNGA432T0420
CNGA433T0420
CNGA434T0420
CNGA643T0420
DNGA544
DNGA432T0420
DNGA433T0420
SNGA433T0420
SNGA434T0420
TNGA432T0420
TNGA433T0420





medium	0
machining	1540
ANSI	
RPG32	•
RPG43	•

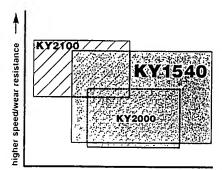
medium machining ANSI RCGV23 RCGV35 RCGV45 RCGV35T0420 RCGV45T0420

i cennamerikon dhe erace KyoN-siajon ceramik ena Indicates uncoated grade Suited for material gijou high-temp alloys ivioleti (orghiese





Kyon 2100 Inserts for High Speed Machining Nickel-Base Alloys



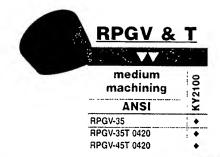
higher feed/toughness



TOOLtips

New process and material technologies derived from grade KY1540 are also being applied to our existing KY2100 sialon ceramic inserts which remain the preferred choice for medium machining nickel-base alloys such as Inconel, Hastelloy, and Waspaloy at high speeds and light to medium feeds.

As well as high thermal and mechanical shock resistance, Kyon 2100 has outstanding resistance to depth-of-cut notching, the main failure mode when machining high-temp alloys.





NG & T

medium machining	KY2100
	~ ~
ANSI	Z
CNG-432T 0820	•
CNG-433T 0420	•
CNG-434T 0420	•
RNG-43	•
RNG-45	•
RNG-43T 0420	
RNG-45T 0420	•
SNG-433	
SNG-434	•
SNG-644	•
SNG-434T 0420	•
SNG-453T 0420	•

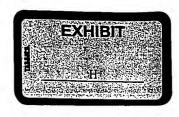


RCGV &

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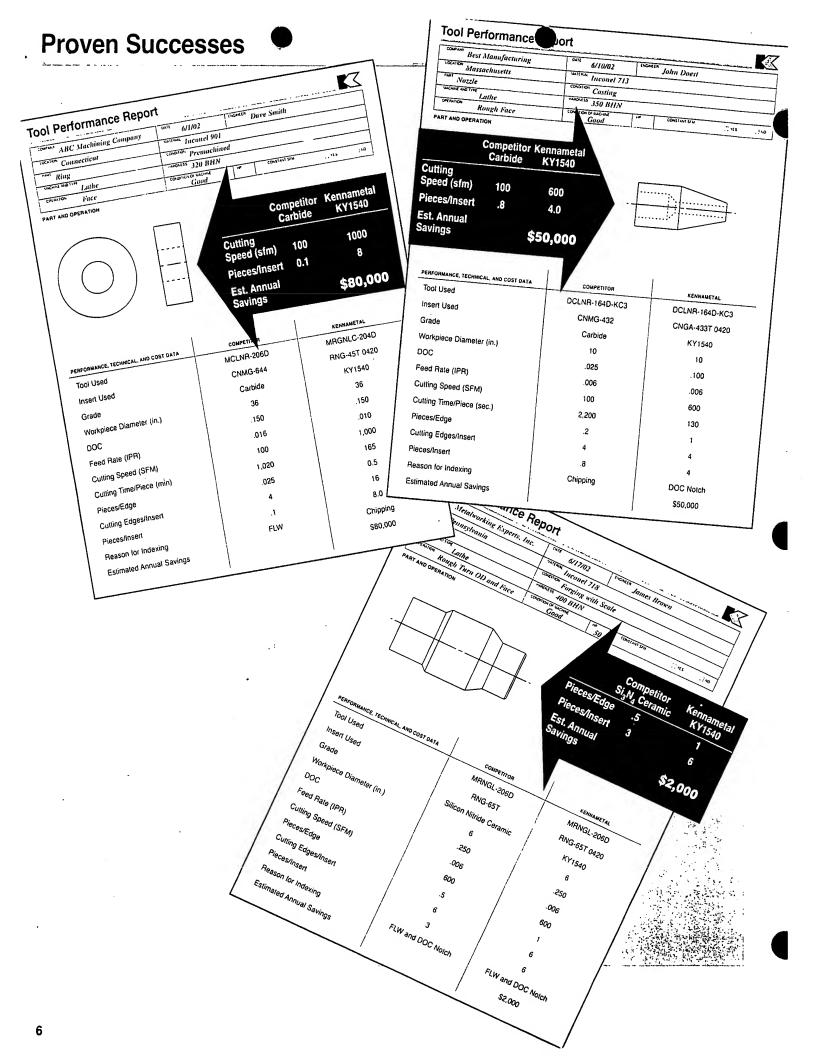
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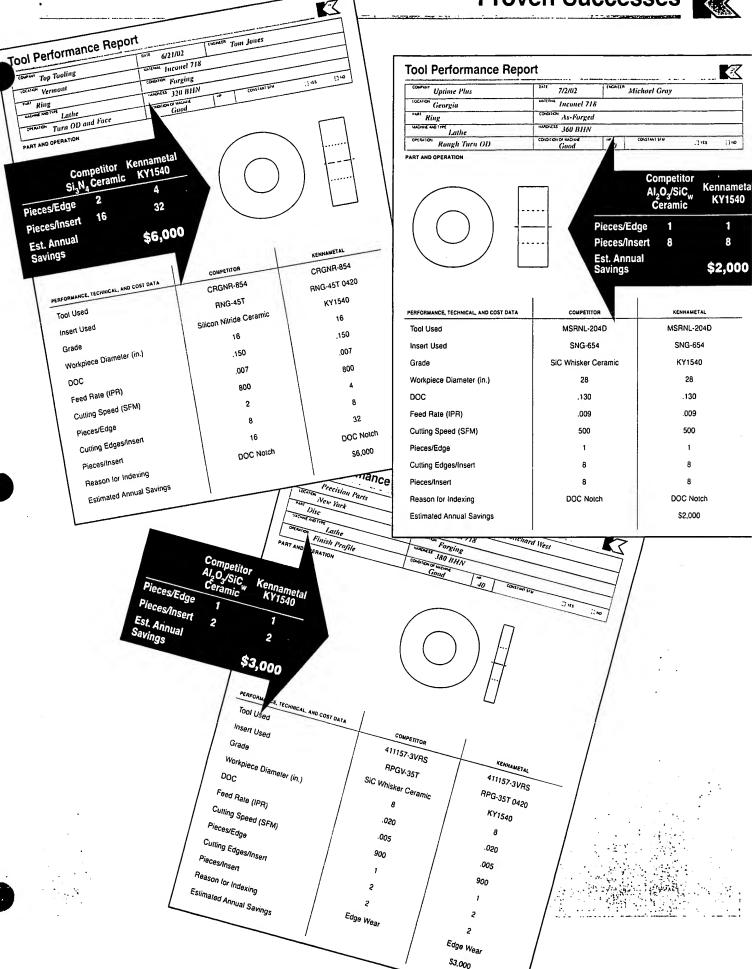
KY1540 Sales (Including as SPM014)

Fiscal		Sales		Sales
Year	Month	Pieces	Dollars	
FY 2002	1	807	\$	8,017
	2	2,447	\$	24,041
	3	856	\$	8,517
	4	2,970	\$	28,578
	5	4,500	\$	44,775
	6	5,600	\$	55,720
	. 7	7,643	\$	80,049
	. 8	6,500	\$	63,957
/	9	10,906	\$	111,021
	10	5,220	\$	54,665
	11	11,564		120,465
	12	5,384	\$	57,586
FY2003	1	2,872		29,177
	2	3,273		35,877
	3	2,254	\$	25,529
	4	2,457	\$	30,811
	5	2,144	\$	25,327
	6	3,363	\$	37,941
	7	3,518	\$	44,363
	8	4,442	\$	52,841
	9	6,287	\$	73,879

95,007 1,013,137









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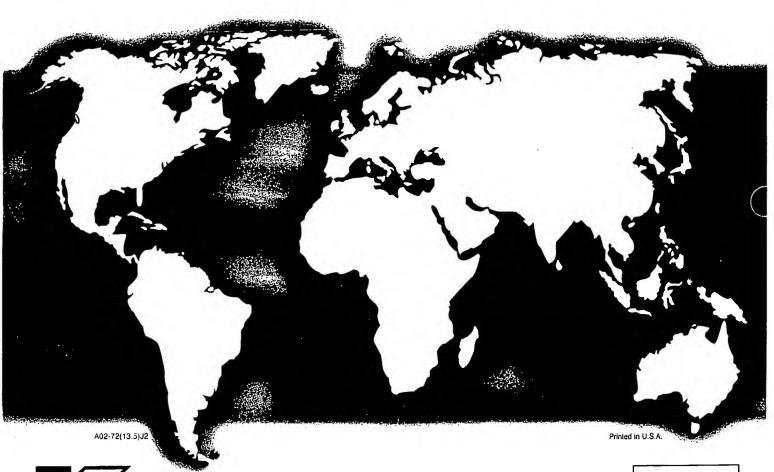
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